



Characteristics of surface passivation of ozone- and water-based Al_2O_3 films grown by atomic layer deposition for silicon solar cells

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ABSTRACT

We investigated the effects of the thermal stability of atomic layer deposition (ALD) oxidants on the surface passivation of ALD- Al_2O_3 film. The results showed good passivation at temperatures not greater than 780 °C. However, we found that Al_2O_3 films with an ozone oxidant showed better surface passivation at high temperatures than the water-based samples. The Al_2O_3 films with a water oxidant yielded an additional interfacial oxide upon high-temperature annealing. In the case of the ozone-based samples, the interfacial Si–O bonds that formed during deposition were more stable. This structural change degraded chemical passivation, which increased the interface-trap density to $\sim 10^{12} \text{ eV}^{-1} \text{ cm}^{-2}$. The passivation performance of ALD- Al_2O_3 films showed that at temperatures over 780 °C the passivation quality was affected more by defective passivation at the Si/SiO_x interface than by a negative-fixed charge.

1. Introduction

Al_2O_3 films grown by atomic layer deposition (ALD) are attractive materials for the surface passivation of high-efficiency crystalline silicon solar cells [1–5]. It is well-known that ALD-grown- Al_2O_3 films' excellent surface passivation on crystalline Si substrates is due to field-effect passivation induced by negative-fixed charge as well as chemical passivation due to low interface defect density [1–5]. Thermal ALD is sensitive to surface conditions and interfacial properties due to the surface reaction of alternative precursor supplies. Therefore, the characteristics of ALD- Al_2O_3 films grown by using H_2O as the oxidant are strongly dependent on the hydroxyl groups [5]. Alternatively, because of ozone's higher activity for ligand elimination in metal-oxide ALD, ozone (O_3)-based ALD- Al_2O_3 films have the advantages of not only minimizing the formation of OH and related defects but also reducing oxygen deficiency [6]. On the other hand, there is not much information on how these two reactants affect the surface-passivation stability of Al_2O_3 , because screen-printed silicon solar cells are subject to a high-temperature firing process for contact to a metal electrode.

In this study, we compare the thermal ALD passivation stabilities of Al_2O_3 for H_2O and O_3 as the primary oxidants.

2. Experimental details

Al_2O_3 films were formed on solar-grade silicon wafers and deposited

by ALD in a NCD Lucida batch-type reactor. All solar-cell samples were fabricated on boron-doped p-type monocrystalline silicon wafers of 1–2 $\Omega\text{-cm}$ resistivity, $180 \pm 20 \mu\text{m}$ thickness, and $156 \text{ mm} \times 156 \text{ mm}$ size.

The Cz Si wafers were cleaned by a standard Radio Corporation of America (RCA) cleaning method of SC-1, SC-2, and HF-dip immediately prior to Al_2O_3 deposition. A 20 nm-thick Al_2O_3 layers were deposited at 250 °C. Trimethylaluminum (TMA) was used as the Al precursor and with either H_2O or O_3 as the oxygen precursor. O_3 was injected into the reactor in a flux 1 SLM at a concentration of 220 g/Nm³ using an MKS AX8560 ozone generator. After ALD Al_2O_3 growth, the wafers were annealed at 450 °C for 30 min in a forming gas ambient (FGA) and fired at 780 °C for 4 s. All temperature readings were obtained using a thermocouple. To examine the thermal stability of the passivation performance on ALD Al_2O_3 passivation layer, the wafers were treated for 4 s at 830 °C, which temperature exceeded the firing temperature by 50 °C.

The carrier lifetimes were measured with Sinton WCT-120 Quasi Steady State Photoconductance equipment at an injection level of $5 \times 10^{15} \text{ cm}^{-3}$. Each measurement was performed thrice and the results were averaged for five samples of surface area $156 \text{ mm} \times 156 \text{ mm}$. The interfacial characteristics were determined by X-ray photoelectron spectroscopy (XPS) and capacitance-voltage (C-V) measurements. The C-V measurements were performed on a MIS capacitor that was formed by the evaporation of a 300 nm aluminum electrode. The interface-defect densities were measured by the conductance method [7].

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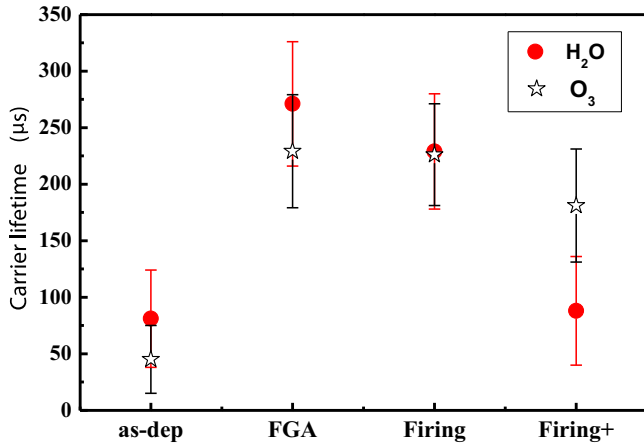
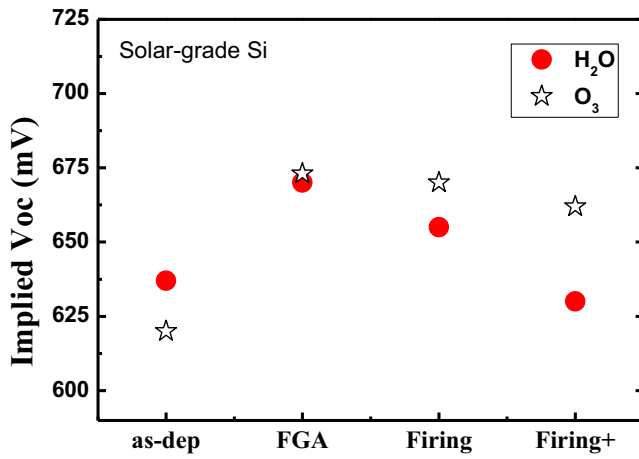
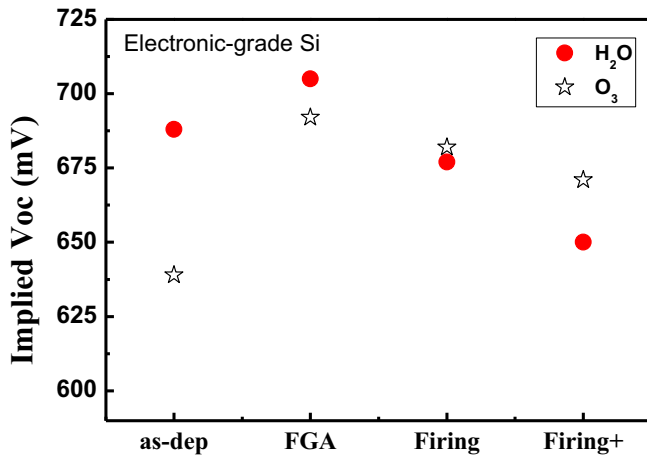


Fig. 1. Carrier lifetimes of ozone- and water-based Al_2O_3 films on solar-grade silicon wafers as functions of the annealing temperature.



(a). H.S. Chang.



(b). H.S. Chang.

Fig. 2. Implied V_{oc} of ozone- and water-based Al_2O_3 films on (a) solar-grade, and (b) electronic-grade silicon wafers as functions of the annealing temperature.

3. Results and discussion

Fig. 1 plots the carrier lifetime as a function of thermal treatment for an ALD Al_2O_3 film using the oxidants H_2O and O_3 . For the as-deposited samples, the water-based Al_2O_3 film showed a longer carrier lifetime

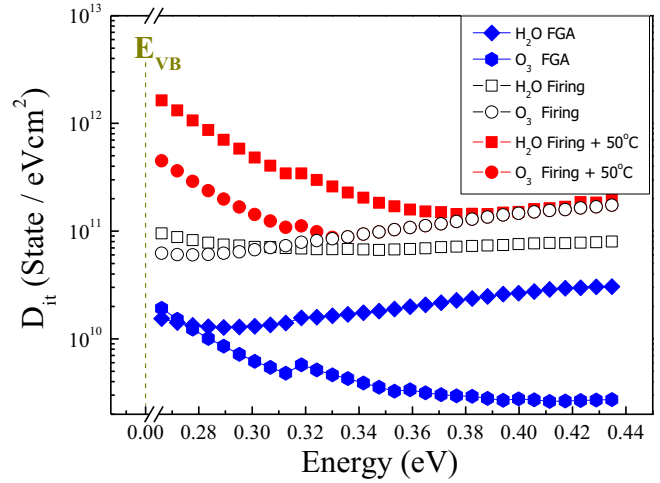


Fig. 3. Interface-trap density as function of energy in the Si band gap, as determined for Al/20 nm Al_2O_3 /p-Si metal-insulator-semiconductors by the conductance method.

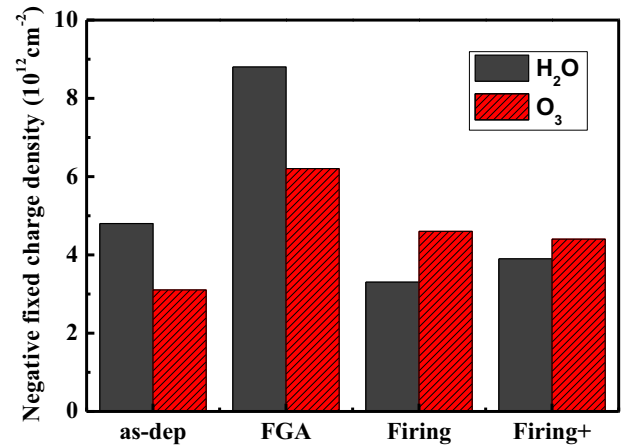


Fig. 4. Negative-fixed-charge density for ozone- and water-based Al_2O_3 films as functions of the annealing temperature.

than the ozone-based film. It is known that FGA treatment imparts the longest carrier lifetime to ALD Al_2O_3 films [8–10]. The average values of carrier lifetime for the water-oxidant and ozone-oxidant samples were 271 and 230 μs , respectively. The average carrier lifetimes after the firing process were similar, on the order of 225–230 μs . There was significant degradation of the lifetime value for the water-based sample after thermal treatment at a temperature 50 $^{\circ}\text{C}$ higher than the firing temperature. The average carrier lifetime for the ozone-based sample was 180 μs . This value, though less than that after firing, still corresponds to a good level of passivation. In other words, the degradation in the passivation of the ozone-based Al_2O_3 layer was less than that of the water-based Al_2O_3 layer after firing at a temperature 50 $^{\circ}\text{C}$ higher (firing + 50 $^{\circ}\text{C}$).

Fig. 2(a) plots the implied open-circuit voltages (V_{oc}) of the H_2O - and O_3 -oxidant Al_2O_3 films as a function of thermal treatment. The water-based films show a better V_{oc} than the ozone-based films, similar to the trend observed with carrier lifetimes. The Al_2O_3 films yielded the best values of V_{oc} : 670–673 mV for the FGA. The ozone-based samples were less degraded after both normal firing and firing at 50 $^{\circ}\text{C}$ higher temperature, as reflected in the V_{oc} values 670 and 662 mV respectively. In constant, the V_{oc} of the water-based film reduced to 655 mV after firing; it drastically reduced to 630 mV after firing 50 $^{\circ}\text{C}$ higher. Though there are differences in the levels of wafer quality, the electronic-grade silicon wafer shows an effect on passivation similar to that on temperature, as shown in Fig. 2(b). There were significant changes

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