



Structural, optical, photoluminescence and photocatalytic investigations on Fe doped TiO₂ thin films

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ARTICLE INFO

Article history:

Received 17 October 2012

Received in revised form 17 October 2013

Accepted 18 October 2013

Available online 30 October 2013

Keywords:

Iron
Titanium dioxide
Sputtering
Micro Strain
Dislocation Density
Band Gap
Photoluminescence
Photocatalytic Activity

ABSTRACT

Pure and Fe doped TiO₂ thin films were deposited onto quartz substrates maintained at room temperature by radio frequency magnetron sputtering. The films were annealed at 873 K in air for 2 h and characterized using X-ray diffraction (XRD), micro Raman spectroscopy, Scanning electron microscopy (SEM), Energy dispersive spectroscopy, ultraviolet-visible and photoluminescence (PL) spectroscopy. Pure TiO₂ thin films were XRD-amorphous, but micro Raman spectra revealed the presence of anatase phase. Doping with Fe has influenced the transformation of films to anatase phase. Improved crystallinity was observed in the 0.1 at% Fe doped films, where the lattice constants approached the values for bulk anatase TiO₂, and the films showed minimum strain. At higher Fe concentrations, micro Raman spectra revealed the presence of rutile phase also. SEM images revealed crack free surface and surface roughness was found to increase with increase in Fe concentration. Doping with Fe has resulted in a red shift of absorption edge. PL emission intensity was found to increase with Fe concentration, but at higher concentrations quenching of PL emission was observed. Fe doping resulted in enhancement of photocatalytic activity, evaluated by monitoring the degradation of methylene blue solution. 0.8 at% Fe doped TiO₂ films exhibited the highest photocatalytic activity.

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1. Introduction

Titanium dioxide (TiO₂) photocatalysts have attracted much interest for air and water purification ever since the discovery of photo-induced water cleavage on TiO₂ electrodes by Fujishima and Honda in the early 1970's. [1]. Dye sensitized solar cells based on TiO₂ is also a subject of interest in the recent years because of their low cost, simple process and environment-friendly photovoltaic ability with good efficiencies compared to conventional silicon solar cells. Also, TiO₂ has long term stability against photo-corrosion and chemical corrosion. Of the two main crystalline polymorphs of TiO₂—rutile and anatase—anatase phase has been widely used as photocatalyst because of its high photo activity. But two factors mainly limit the use of TiO₂ thin films in photocatalytic process: (1) Because of its high band gap, outstanding electronic properties and chemical stability, it can be photoactivated only in the presence of ultraviolet (UV) light, which constitutes less than 5% of total solar energy. Doping with transition metals and non-metallic impurities is a promising route to solve this problem. (2) Electron-hole pairs created undergo very fast recombination, releasing energy in the form of unproductive heat. Hence recombination rate of charge carriers has to be reduced by providing suitable trap states.

To improve the photoactivity of TiO₂, so that absorption threshold can be extended to visible range, transition metals such as Fe, V, Mn, Co and Ni [2–7] can be used as dopants. The transition metal ions can create intermediate energy levels into the band gap of TiO₂. This causes a narrowing of band gap which enables visible light itself to initiate photoactivity. Of these various transition metals Fe is advantageous because of two reasons: (1) the ionic radius of Fe³⁺ (0.064 nm), is very close to that of TiO₂ (0.068 nm) so that Fe³⁺ ions can easily get incorporated into the crystal lattice of TiO₂ and (2) Fe³⁺ ions can act as electron-hole traps and prevent their recombination as the energy level of Fe²⁺/Fe³⁺ lies close to that of Ti³⁺/Ti⁴⁺ [8].

Iron doped TiO₂ (Fe–TiO₂) thin films have been prepared by a variety of techniques such as sol–gel method [9,10], mechanical alloying [11], co-precipitation [12], hydrothermal method [13], controlled hydrolysis [14], metal organic chemical vapor deposition [15] and reactive magnetron sputtering [16,17]. There are varied reports in literature that Fe doped TiO₂ thin films can either increase [12,14,18–21] or decrease [9,22–25] the photocatalytic activity. Also there are diverse reports on the structural patterns such as phase and microstructure of Fe–TiO₂ thin films [26–28]. This has prompted us to undertake further studies on Fe doped TiO₂ thin films. Also radio frequency (RF) magnetron sputtering has the advantages of obtaining uniform large area coatings which have good adherence to the substrate. The present study reports on the structural, optical, photoluminescence and photocatalytic

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properties of Fe–TiO₂ thin films deposited on quartz substrates by RF magnetron sputtering.

2. Experimental details

2.1. Preparation of thin films

Pure TiO₂ and Fe₂O₃ (99% Sigma Aldrich) powders were used for target preparation. They were mixed together in the desired proportions and ball milled for 8 h using ball milling apparatus. The sputtering target was prepared by standard ceramic procedure [29]. Six different targets were prepared by changing iron concentration. Fe/Ti atomic ratio in the individual targets was 0, 0.05, 0.1, 0.5, 0.8 and 1.1 at%. The substrate–target distance was 5 cm. Sputtering was carried out at sputtering pressure 1.5 Pa and RF power 120 W for 2 h in pure Ar atmosphere. The films were coated on quartz substrates initially maintained at room temperature. During sputtering, the substrate temperature increased up to 353 K. Post deposition thermal annealing of the films at 873 K was done in air for 2 h. The films thus obtained were coded TO[873], 0.05FTO[873], 0.1FTO[873], 0.5FTO[873], 0.8FTO[873] and 1.1FTO[873] respectively.

2.2. Characterization of thin films

X-ray diffraction (XRD) patterns of the films were recorded using Bruker D8 Advance X-ray diffractometer using Cu K α radiation ($\lambda = 0.15405$ nm) at a scan rate of 0.05°s^{-1} . Bragg–Brentano θ – θ configuration was used for the scan. Scanning range was 20 to 60° , and scanning step size was 0.01° . The accelerating voltage and applied current were 40 kV and 40 mA, respectively. The average crystallite size of the film was calculated using Debye–Scherrer formula [30]. The lattice parameters and microstrain of the films were also evaluated [31]. Micro Raman analysis of the films was done using a Labram-HR 800 spectrometer equipped with an argon ion laser. Scanning electron microscopy (SEM) images were recorded using JEOL Model JSM-6390LV microscope equipped with a Energy dispersive spectroscopy (EDX) set up (JEOL Model JED-2300). The accelerating voltage was in the range 20–30 kV for the various films. For EDX measurements, high vacuum resolution of 3 nm was used. Ultraviolet–visible (UV–vis) transmission spectra of the films were recorded using JASCO V-550 spectrophotometer. From the transmission spectra, Swanepoel's envelope method [32] was used to calculate the optical constants and Tauc plot [33] was used to determine the optical band gap of the films. Porosity values were calculated using the relation [34]:

$$P = 1 - \frac{n^2 - 1}{n_d^2 - 1} (\times 100\%)$$

where P is the porosity, n the refractive index of TiO₂ thin film and n_d that of bulk TiO₂. Photoluminescence (PL) spectra of the samples were recorded using PerkinElmer LS 55 spectrofluorometer equipped with a 40 W Xenon lamp. Excitation wavelength used was 320 nm.

2.3. Evaluation of photocatalytic activity of thin films

The photoreactor system used in the present study was a rectangular quartz cell ($80 \times 50 \times 3$ mm). 20 ml of 0.01 g/l methylene blue (MB) solution was taken in the cell. TiO₂ thin film of dimension $50 \times 25 \times 1.35$ mm was suspended in this solution. The solution was circulated with air during the reaction. The system was irradiated with light from a 125 W high pressure mercury vapor lamp. The irradiation density was 100 mW/cm^2 . The photocatalytic experimental setup was kept in dark and the solution was stirred continuously for 30 minutes to ensure adsorption/desorption equilibrium, before irradiation with mercury lamp. This equilibrium concentration was used as the initial value for further photodecomposition process. The absorbance of

solution was measured after every half hour, for 2 h. MB has a maximum absorbance at 663 nm. The change in MB concentration with irradiation time was determined by a UV–vis spectrophotometer by measuring its absorbance at 663 nm. Percentage degradation (D) of MB solution was evaluated as follows: $D = (A_0 - A) / A_0 \times 100$ where A_0 is the initial absorbance of the dye solution and A , its absorbance at different irradiation times. The apparent rate constant was also evaluated [35].

3. Results and discussions

3.1. Structural studies

Crystalline structure of pure TiO₂ and Fe–TiO₂ thin films were examined by XRD and micro Raman spectroscopy. Analysis of the XRD and micro Raman spectra are given below:

3.1.1. XRD Studies

Fig. 1 shows the XRD patterns of pure TiO₂ and Fe–TiO₂ thin films post-annealed at 873 K. Pure TiO₂ thin films were XRD-amorphous. But as Fe content was increased, the films exhibited crystallization to anatase (≤ 0.1 at%) and then to mixed phase (0.5 at%). For the film 0.05FTO[873], two diffraction peaks at $2\theta = 25.3919$ and 37.7948° corresponding to reflections from A(101) and A(004) planes were observed. Four diffraction peaks at $2\theta = 25.2607, 37.7350, 53.7484$ and 55.1023° were observed, corresponding to reflections from A(101), A(004), A(105) and A(211) planes for the film 0.1FTO[873]. In addition to reflections from A(101), A(004), A(200) and A(105) planes at $2\theta = 25.2837, 37.9356, 48.1654$ and 53.6815° respectively, a weak peak corresponding to reflection from R(110) plane at $2\theta = 27.7726^\circ$ was observed in the film 0.5FTO[873]. The 0.8 and 1.1 at% doped Fe–TiO₂ thin films were again amorphous-like in nature. A weak peak corresponding to reflection from A(101) plane was observed at $2\theta = 25.9063^\circ$ for the film 0.8FTO[873]. No preferential orientation along a particular plane was noted in the investigated films. Also, diffraction peaks corresponding to oxides of iron such as FeO, Fe₂O₃, Fe₃O₄ or any mixed oxides such as Fe_xTiO_{2y} were not observed in the XRD patterns. This can be attributed to the fact that most of the Fe ions were inserted into the TiO₂ lattice sites [15].

Also, the intensity of diffraction peaks was found to increase with increasing Fe dopant concentration up to 0.5 at%. This indicates an increase in film crystallinity [28]. But at higher doping concentrations (≥ 0.5 at%), the films appear amorphous which means crystallinity is

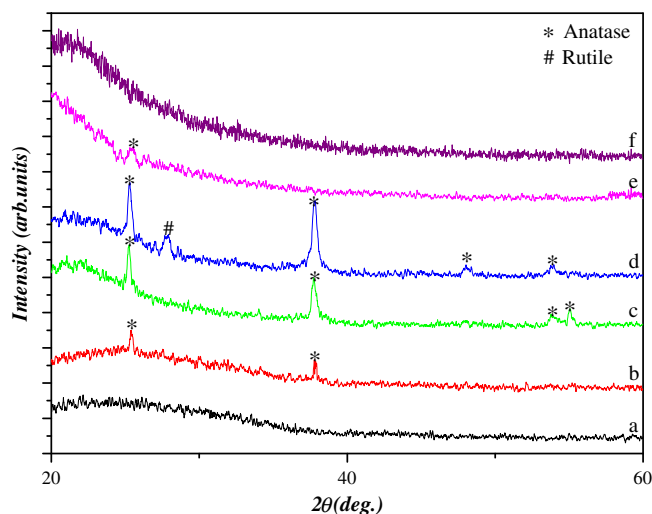


Fig. 1. XRD patterns of (a) pure TiO₂ and (b) 0.05 (c) 0.1 (d) 0.5 (e) 0.8 (f) 1.1 at % Fe–TiO₂ thin films post-annealed at 873 K.

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