



# Investigation of structural and magnetic properties of nanocrystalline Mn-doped $\text{SrFe}_{12}\text{O}_{19}$ prepared by proteic sol–gel process



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## ARTICLE INFO

### Article history:

Received 22 May 2015

Received in revised form

21 July 2015

Accepted 25 July 2015

Available online 28 July 2015

### Keywords:

M-type hexaferrite

Proteic sol–gel process

Crystal structure

Magnetic property

## ABSTRACT

Nanoparticles of  $\text{SrFe}_{12-x}\text{Mn}_x\text{O}_{19}$  ( $x=0.0$  and  $0.10$ ) were synthesized by a proteic sol–gel process. Thermogravimetric and differential thermal analyses (TG–DTA) indicated the formation of nanocrystalline strontium ferrite phase at a calcination temperature of  $1000^\circ\text{C}$ . Structural and microstructural evolutions of the samples were studied by X-ray powder diffraction (XRD) and the Rietveld method. XRD patterns demonstrated that all samples consisted of single-phase M-type strontium hexaferrite. The crystal lattice constant did not change significantly with manganese substitution, ranging from  $0.5877(3)$  nm ( $x=0.0$ ) to  $0.5876(3)$  nm ( $x=1.0$ ). In addition, the average crystallite size, which was determined from the Williamson–Hall formula, was about  $46.4$ – $52.6$  nm. Infrared spectroscopy (FT-IR) showed the presence of three principal absorption bands in the frequency ranges around  $435$ – $535\text{ cm}^{-1}$  and around  $595\text{ cm}^{-1}$ , indicating the formation of the hexaferrite. Scanning electron microscopy (SEM) revealed that particles consisted of irregular platelets with sizes from  $68$  to  $204$  nm. Room-temperature Mössbauer investigations revealed that manganese ions preferentially occupied the  $12k$ ,  $4f_1$ ,  $4f_2$ , and  $2a$  sites. Hysteresis loops (M–H) showed that the saturation magnetization, remanence, and coercivity decreased with manganese doping. This effect is discussed in terms of the distribution of metal cations in the tetrahedral and octahedral sites.

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## 1. Introduction

In recent years, M-type strontium hexaferrite (Sr-hexaferrite,  $\text{SrFe}_{12}\text{O}_{19}$ , SrM) has been the subject of extensive research due to its possible use in different technological applications. These materials are used in the magnetic recording industry and in magneto-optic devices, microwave devices, microelectromechanical systems, transformer cores, and antennas [1].

The physical properties of strontium ferrite depend intrinsically on the synthesis route. Strontium hexaferrite is a ferromagnetic material having a structure belonging to the hexagonal space group  $P6_3/mmc$  and a magnetocrystalline anisotropy parallel to the  $c$ -axis of the unit cell. The crystalline structure of Sr-hexaferrite is constructed from a periodic repetition of  $\text{SRS}^*\text{R}^*$  blocks, where the cubic S block has a spinel structure and the hexagonal R block contains a  $\text{Sr}^{2+}$  ion, and the  $\text{S}^*$  and  $\text{R}^*$  blocks are axially symmetric around the hexagonal axis- $c$  by  $180^\circ$ . Twenty-four  $\text{Fe}^{3+}$

ions occupy interstitial positions in five different crystallographic sites and are distributed as follows. Three octahedral sites ( $12k$ ,  $4f_2$ , and  $2a$ ) and site  $4f_1$  have a geometric tetrahedron formed by oxygen atoms. Site  $2b$  has a bipyramidal geometry with a hexahedral triangular base formed by five oxygen atoms around the  $\text{Fe}^{3+}$  ion [2]. Numerous researchers [3,4] have been working on the effect of substitution of  $\text{Fe}^{3+}$  ions with different compounds, where the partial replacement of these ions modifies various magnetic properties [5,6].

Herein, we present a study of the effect of substitution of  $\text{Fe}^{3+}$  by  $\text{Mn}^{3+}$  on the structural and magnetic properties of strontium hexaferrite synthesized by the proteic sol–gel process.

## 2. Experimental

### 2.1. Synthesis

M-type  $\text{SrFe}_{12-x}\text{Mn}_x\text{O}_{19}$  ( $x=0$  and  $0.10$ ) samples were synthesized by the proteic sol–gel process. Appropriate amounts of  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  (99% Sigma-Aldrich),  $\text{Sr}(\text{NO}_3)_2$  (99% Sigma-Aldrich),

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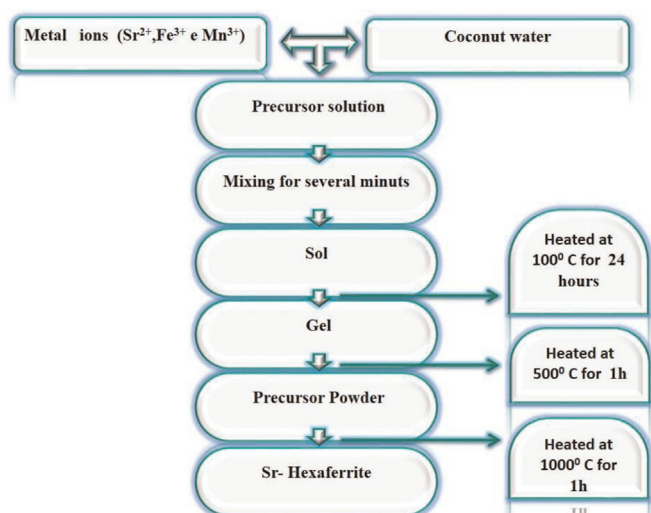


Fig. 1. Flow chart for sample preparation of Sr-hexaferrite nanopowders by the proteic sol-gel process.

and  $\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$  (99% Sigma-Aldrich) were dissolved in filtered coconut water and mixed for several seconds to form the sol. The sol was dried at  $100^\circ\text{C}$  for 24 h for gel formation and dehydration. Afterwards, it was transformed into xerogel. For the decomposition of organic materials and salts, the xerogel was calcined first at  $500^\circ\text{C}$  for 1 h at a heating rate of  $1^\circ\text{C}$  per minute in an open atmosphere. Then, the temperature was increased at a rate of  $1^\circ\text{C}$  per minute up to a temperature of  $1000^\circ\text{C}$ , where it remained for 1 h, and then the xerogel was removed from the oven at room temperature. The systematic procedure for the synthesis of hexaferrite is illustrated in Fig. 1.

## 2.2. Characterization

The thermal behavior of the samples was studied by thermogravimetric and differential thermal analyses (TG–DTA) using a simultaneous TG–DTA (TA Instruments) at a heating rate of  $10^\circ\text{C}$  per minute in air synthetic with a flux of 100 ml/min. Fourier transform infrared (FT-IR) spectroscopy measurements of the samples were performed on a Virian Resolutions Pro in the range of  $400\text{--}4000\text{ cm}^{-1}$  with KBr pellets. X-ray diffraction (XRD) analysis of the calcined  $\text{SrFe}_{12-x}\text{Mn}_x\text{O}_{19}$  (with  $x=0.0$  to 0.10) samples was performed in a Rigaku X-ray diffractometer using a  $\text{Cu K}\alpha$  radiation tube operated at 40 kV and 40 mA. XRD data were taken in step scan mode in the range of  $20\text{--}80^\circ (2\theta)$  with step sizes of  $0.02^\circ$  and a scan speed of  $0.02^\circ/\text{min}$ . The structural and microstructural parameters were extracted using Rietveld refinement with the program FullProf [7]. The Bragg peaks were modeled with the pseudo-Voigt function, and the background was estimated by linear interpolation between selected background points. The determination of the average crystallite sizes ( $D$ ) and the lattice microstrain ( $\epsilon$ ) was estimated by line-broadening analysis using Williamson–Hall (W–H) [8] methods. The morphology, microstructure, and particle size of the  $\text{SrFe}_{12}\text{O}_{19}$  and  $\text{SrFe}_{11.9}\text{Mn}_{0.10}\text{O}_{19}$  nanopowders were examined by scanning electron microscopy (SEM) using a MIRAN 3 TESCAN with an accelerating voltage of 13.0 kV. The SEM images were analyzed with the software ImageJ [9]. The magnetic properties of SrM samples were measured at room temperature using a maximum external field of 70 kOe and a Magnetic Properties Measurements System (MPMS; Quantum Design).  $^{57}\text{Fe}$  Mössbauer effect measurements were taken at room temperature in a transmission geometry using a conventional constant-acceleration spectrometer operating in triangular wave mode with a  $^{57}\text{Co}$  source in a rhodium matrix. The spectrometer

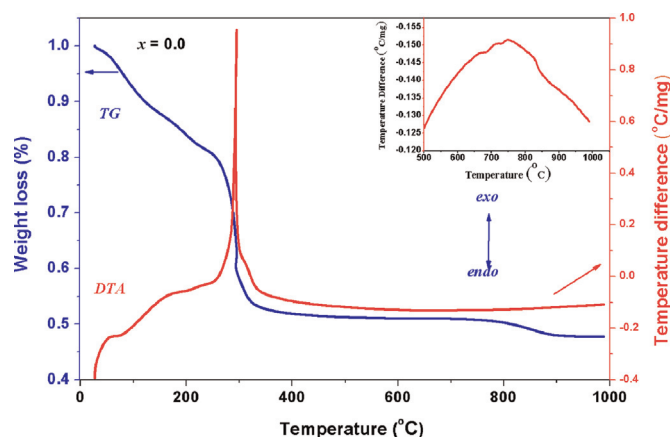


Fig. 2. Thermogravimetric (TGA) curve and differential thermal analysis (DTA) curve of the sol-gel precursor,  $\text{SrFe}_{12-x}\text{Mn}_x\text{O}_{19}$  ( $x=0.0$ ).

was calibrated with a  $25\text{-}\mu\text{m}$ -thick  $\alpha\text{-Fe}$  foil.

## 3. Results and discussion

### 3.1. Thermal analysis

Fig. 2 shows the TG–DTA curve of the synthesized  $\text{SrFe}_{12-x}\text{Mn}_x\text{O}_{19}$  ( $x=0.0$ ). Thermal analysis of the sample was carried out up to  $1000^\circ\text{C}$  at a heating rate of  $10^\circ\text{C}/\text{min}$  in an argon gas with platinum crucibles to observe the structural variations, such as the weight loss and transformation of different phases during heat treatment. The TGA curve shows that the first mass loss occurred between  $45^\circ\text{C}$  and  $100^\circ\text{C}$  along with an exothermic peak at  $53^\circ\text{C}$  and a slightly weak endothermic trough at  $76^\circ\text{C}$  in the DTA curve, which are respectively associated with the decomposition of nitrates and the evaporation of both water molecules still present in the sample and water adsorbed from the atmosphere after the preparation of the xerogel before measurement. The second mass elimination between  $150$  and  $258^\circ\text{C}$  is associated with the melting of proteins [10]. At temperatures of  $272$  and  $334^\circ\text{C}$ , it was possible to observe a third mass loss associated with an exothermic peak at  $293^\circ\text{C}$  due to the probable formation of hydroxides. The fourth mass loss between  $370$  and  $783^\circ\text{C}$  is correlated with the decomposition of hydroxides and organic matter from the precursor agents in coconut water as well as the formation of several metal oxides and mono-ferrite. The fifth exothermic peak at  $754^\circ\text{C}$  suggests the formation of the hexaferrite phase (inset in Fig. 2) [11].

### 3.2. FT-IR spectroscopy

The spectra were recorded by mixing 0.05% in KBr powder to obtain better resolution of the bands. Fig. 3 shows the absorption bands at  $435$ ,  $535$ , and  $595\text{ cm}^{-1}$ , which are characteristic of hexaferrite [12,13]. The peaks at  $1200$ ,  $1470$ , and  $1560\text{ cm}^{-1}$  are related to the M–O–M (metal–oxygen–metal) bands, such as Fe–O–Mn and Fe–O–Fe. The presence of these bands suggests that Fe–O and Sr–O bonds could interact with Mn dopant atoms. The variation in the band positions resulted from differences in the distances between tetrahedral and octahedral sites at which they occurred in the intervals of  $400\text{--}500\text{ cm}^{-1}$  and  $500\text{--}700\text{ cm}^{-1}$ . This directly resulted in a shift of the absorption peaks due to variations in the distances of  $\text{Fe}^{3+}\text{--O}^{2-}$  caused by the replacement of  $\text{Mn}^{3+}$  [14]. The relatively broad peaks at  $3440\text{ cm}^{-1}$  for both samples became narrow and are associated with the deformation vibration of hydroxyl groups (OH) obtained in a wet atmosphere.

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