

# Structural investigations of CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glasses by Raman spectroscopy and XPS considering its application to continuous casting of steels



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## ARTICLE INFO

### Article history:

Received 23 January 2015

Revised 27 February 2015

Accepted 21 March 2015

Available online 21 March 2015

### Keywords:

Oxy-fluoro-nitride glass

Viscosity

Shear thinning

Degree of polymerization

Mold flux

## ABSTRACT

Structural analysis of CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass was conducted in order to understand its viscoelastic properties. CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass has been adapted in various industries because a small amount of nitrogen incorporation leads to enhancing many properties such as fracture toughness, viscosity and chemical durability. Recently, few metallurgists have begun to use CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass as a lubricant of continuous casting process of steel in order to maximize its viscoelastic property by controlling degree of polymerization (DP). The present study provides structural understanding of CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass, specifically designed for the continuous casting process of steel by Raman spectroscopy and XPS. With Raman spectroscopy, the degree of polymerization (DP) was evaluated by the  $Q_3/Q_2$  ratio. The qualitative analysis of nitrogen and non-bridging oxygen was conducted by XPS within a range between 0.1 and 6 wt.% of nitrogen. According to the results, the introduced nitrogen was incorporated with non-bridging oxygen of silicate network units, resulting in increase on degree polymerization (DP) of silicate network units. Such structural changes would potentially enhance shear thinning property of CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass, resulting in a positive influence on continuous casting process of steel.

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## 1. Introduction

CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass are amorphous solids derived from the silica structure by replacing part of the bivalent oxygen atoms by trivalent nitrogen atoms in a way that silicon is bonded simultaneously to oxygen and nitrogen atoms [1]. Due to the such superiority in bonding number, it is well known that replacement of small amounts of nitrogen in the glass by oxygen causes considerable changes in properties such as glass transition temperature, fracture toughness, viscosity, chemical durability and Young's modulus [2].

Such oxy-nitride glass, in particular, is attractive for glass fiber due to its outstanding electrical conductivity [3]. Apart from glass fiber industry, oxy-nitride glass has high potential applications in jointing or coating materials due to its high toughness and viscosity [3]. Especially, it is interestingly reported that oxy-nitride glass

could be used for the containment of high level nuclear waste for their ultimate disposal due to its good chemical durability [3].

Recently, a rheological property of CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass has been received attention in the steel industry. During the continuous casting process of steel, an amorphous liquid phase of synthetic oxy-fluoride glass or so called mold flux (MF) is sandwiched between oscillating water cooled copper mold and molten steel in order to lubricate the molten steel with achieving successful solidification of steel [4]. In this time, one of the important challenges to overcome is to design MF with a proper value of viscosity [4]. This is because the viscosity of MF has to be large enough at mold top surface (10–40 cm<sup>−1</sup>) in order to alleviate slag entrainment and low enough at mold wall (100–1000 cm<sup>−1</sup>) for maximizing lubricating capability [4].

Under such circumstances, metallurgists begun to study on rheological property of MF in order to design MF having Non-Newtonian behavior of shear thinning property in that viscosity decreases as a function of shear rate. With respect to Non-Newtonian behavior of MF, the author previously reported that MF itself shows a little bit of Non-Newtonian behavior of shear thinning property [4]. It was further suggested that the shear

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thinning property of MF is strongly dependent on DP [4]. Also, Watanabe et al. experimentally proved that nitrogen incorporation induce strong Non-Newtonian behavior of MF [5].

Albeit the previous researches by metallurgists including the author made huge progress in that they successfully achieved stronger shear thinning property with CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass and discovered correlation between shear thinning property and structure of MF, questions still exist concerning MF structure incorporated with nitrogen (CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass) because, as insisted by many researchers, the properties of glass are a function of structural features [6]. Therefore, the structural details of MF incorporated with nitrogen are absolutely needed in order to design MF with strong shear thinning property.

It is the purpose of this study to investigate the structural features of MF incorporated by nitrogen and to design CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass for improving shear thinning property. With Raman spectroscopy, DP of MF incorporated with nitrogen is evaluated by analyzing silicon discrete anion ( $Q_n$ ) units. Subsequently, XPS analysis was conducted in order to understand quantitative features on nitrogen (N1, N2 and N3) and non-bridging oxygen. By combining the results from Raman and XPS, it should be able to understand the effect of nitrogen incorporation on structures of MF.

## 2. Experimental

### 2.1. Preparation of CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass

There are mainly two methods to prepare CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub>. The first and widely used method is the direct melting of all ingredients and nitrides under Ar atmosphere at high temperature [3]. The second way includes the preparation of the base oxide glass and followed by nitridation ammonolysis at high temperature [3]. For the present study, the second method is adapted in order to eliminate the feasibility of Si<sub>3</sub>N<sub>4</sub> decomposition. Rajaram and Day reported that decomposition of Si<sub>3</sub>N<sub>4</sub> is occurred due to the reaction of some gaseous product during melting a single step [7]. The flow chart for synthesis of silicon oxy-nitride glass for the present study is shown in Fig. 1. For the synthesis of CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based

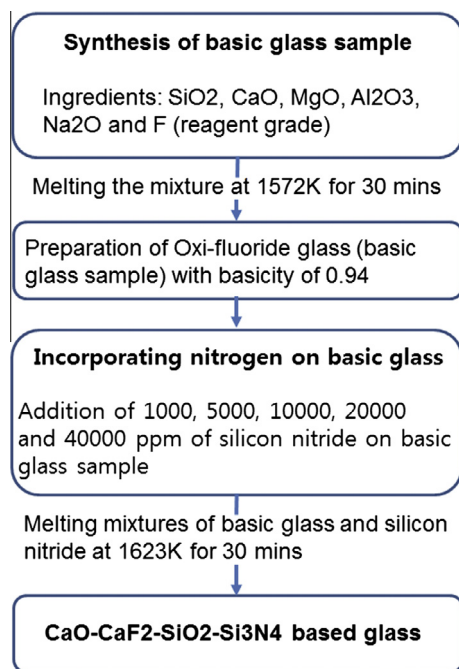


Fig. 1. Flow diagram showing the synthesis of silicon oxynitride glass.

glass, firstly, oxy-fluoride glass with basicity(CaO/SiO<sub>2</sub>) of 0.94 was prepared by melting a mixture, consisting of reagent grade SiO<sub>2</sub>, CaO, MgO, Al<sub>2</sub>O<sub>3</sub>, Na<sub>2</sub>O and F in a platinum crucible at 1572 K for 30 min to obtain homogeneous mixtures. Subsequently, molten mixtures in platinum crucible was quickly taken out from the furnace and followed by a rapid quenching process with water cooled steel mold. The chemical composition of oxy-fluoride glass was designed with a consideration of chemical composition for commercial MF. The specific chemical compositions are analyzed by using X-ray fluorescence (XRF) spectroscopy. For quantification of chemical composition with XRF, a powder form of each sample was subjected to be melted with fusing agent (Di-Lithium tetraborate) at 1723 K and followed by quenching process. The quenched sample was grinded and mechanically mixed in order to achieve homogeneous condition. The prepared powder form of sample was tested by XRF (XRF-1800 model Shimadzu) to determine chemical composition.

After synthesizing oxy-fluoride glass, nitrogen is added to the precursor oxy-fluoride glass as a form of silicon nitride (Si<sub>3</sub>N<sub>4</sub>) at 1623 K for 30 min and followed by the quenching process. The additive silicon nitride amounts were between 0.11 and 6.57 wt.% respectively. The specific amounts of nitrogen in CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass were measured by N/O analysis (HORIBA Model: EMGA-930, Japan). The standard used for quantitative analysis of N was nitrogen (JSS370-1). All samples were crushed by vise and contained in a capsule. For each sample, 5 measurements were conducted to estimate the nitrogen homogeneity and average nitrogen content in the samples. The detailed chemical composition and analyzed nitrogen amounts are provided in Table 1.

### 2.2. Raman spectroscopy

The previously prepared CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass was analyzed by Raman spectroscopy (LabRaman High Resolution, Horiba Jobin-Yvon, France) in order to understand the structure of silicate discrete anions ( $Q_n$ ) units. Studying on the structure of oxy-nitride units with quenched sample could be valid because Mysen and Richet experimentally proves that the structure of anionic unit is same before and after the quenching process [8]. Therefore, for the present study, Raman spectra of quenched CaO–CaF<sub>2</sub>–SiO<sub>2</sub>–Si<sub>3</sub>N<sub>4</sub> based glass samples were collected at room temperature in the range of 200–1600 cm<sup>-1</sup> with an Ar excitation laser source with a wavelength of 514.32 nm for 120 s. After collecting Raman spectra data, the Raman bands ranged between 800 and 1200 cm<sup>-1</sup> were fitted by Gaussian function with PeakFit program version 4 within ±0.5 percent error. The relative amount of silicate discrete anions ( $Q_n$ ) was measured by the area ratio of the best fitted Gaussian curves. As for the each silicate discrete anions, the bands highlighted at 855, 905, 964 and 1056 cm<sup>-1</sup> correspond to  $Q_0$ ,  $Q_1$ ,  $Q_2$  and  $Q_3$  units respectively. The characteristic Raman assignments were listed in Table 2.

Table 1

Chemical composition of silicon oxynitride glass measured by XPS. Foundation stands for the basic silicon oxyfluoride glass for the continuous casting process. The samples labeled from N\_1 to N\_5 are prepared by synthesizing nitrogen with the foundation. The nitrogen contents were analyzed by Horiba N/O analysis.

Sample	SiO <sub>2</sub>	CaO	MgO	Al <sub>2</sub> O <sub>3</sub>	Na <sub>2</sub> O	CaF <sub>2</sub>	N
Foundation Sample	40.1	40.2	1	4.9	7.1	7.5	0
	Nitrogen contents (wt.%)						
N_1			0.110				
N_2			0.371				
N_3			1.00				
N_4			2.45				
N_5			6.57				

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