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# Luminescence properties of a novel orange–red CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub>:Eu<sup>3+</sup> phosphor for near-UV pumped W-LEDs

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#### ABSTRACT

A new orange–red CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub>:Eu<sup>3+</sup> phosphor was synthesized by conventional high temperature solidstate reaction method. Its crystal structure, luminescence properties were investigated. This phosphor can be effectively excited by 394 nm near ultraviolet light and exhibit bright orange–red emission with the emission peaks located at 581, 591 and 614 nm corresponding to the  ${}^{5}D_{0} \rightarrow {}^{7}F_{j}$  (j=0–2) transition of Eu<sup>3+</sup> ions, respectively. The optimum Eu<sup>3+</sup> ions concentration in CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> is 17.0 mol% and the critical transfer distance is determined to be 15.13 Å. Moreover, the CIE chromaticity coordinate and the color purity performances of CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub>:Eu<sup>3+</sup> were also discussed. The present work suggests that the CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub>:Eu<sup>3+</sup> phosphor is a potential orange–red material for the application in the near-UV pumped white light emitting diodes.

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#### 1. Introduction

In recent years, white light emitting diodes (W-LEDs) have been considered as a promising technology for next generation solidstate lighting systems due to their unmatchable advantages, such as high brightness, low power consumption, longer lifetime, and high color rendering index (CRI) [1–3]. The commercial way to approach white light is by using the blue InGaN chip combined with a broadband yellow-emitting  $Y_3Al_5O_{12}:Ce^{3+}$  phosphor [4–6]. However, this method results in low CRI and high correlated color temperature (CCT) due to the deficiency of red emission in the visible spectrum. To solve this problem, an alternative way has been proposed is based on a combination of near ultraviolet (near-UV) LEDs with tri-color (red, green and blue) phosphors [5,7,8]. For example, BaMgAl<sub>10</sub>O<sub>17</sub>:Eu<sup>2+</sup>, is for blue component, (Sr, Ba)<sub>2</sub>SiO<sub>4</sub>:Eu<sup>2+</sup> is for green component and CaAlSiN<sub>3</sub>:Eu<sup>2+</sup> is for red component [9–11]. Therefore, it is meaningful to develop a new kind of phosphor with high performance for near-UV based W-LEDs applications.

Borates, which generally feature the advantages of a low synthesis temperature, excellent chemical and physical stability, have been attracted much attention and widely used as luminescent material hosts [1,12–14]. Such as Ba<sub>2</sub>Ca(BO<sub>3</sub>)<sub>2</sub>:Ce<sup>3+</sup>, NaCaBO<sub>3</sub>:Eu<sup>3+</sup> and Sr<sub>2</sub>B<sub>2</sub>O<sub>5</sub>:Eu<sup>2+</sup> [12,15,16]. Hence, it is a good approach to

http://dx.doi.org/10.1016/j.ijleo.2016.01.155 0030-4026/© 2016 Elsevier GmbH. All rights reserved. develop a novel borate-based phosphor which can be effectively excited by near-UV light. In 2008, a new borate compound CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> has been reported[17]. However, to the best of our knowledge, there is no investigation has been performed on studying the luminescent properties of Eu<sup>3+</sup> doped CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> phosphors.

In this paper, we report a new orange-red phosphor  $CaBi_2B_2O_7:Eu^{3+}$  obtained by solid state reaction method. Its crystal structure, luminescence properties of  $CaBi_2B_2O_7:Eu^{3+}$  were investigated. Moreover, the CIE chromaticity coordinate and the color purity performances of  $CaBi_2B_2O_7:Eu^{3+}$  were also studied. The results suggest that the  $CaBi_2B_2O_7:Eu^{3+}$  phosphor is a potential orange-red material for the application in the near-UV pumped W-LEDs.

#### 2. Experimental

All samples were prepared by conventional solid state reaction method at a high temperature. We used CaCO<sub>3</sub> (AR), H<sub>3</sub>BO<sub>3</sub> (AR), Bi<sub>2</sub>O<sub>3</sub> (AR), Eu<sub>2</sub>O<sub>3</sub> (AR) as starting materials. The stoichiometric materials were weighted and ground together in an agate mortar. Thereafter, the mixture was transferred to a corundum crucible and precalcined at 500 °C for 1 h, and then subsequently further sintered at 800 °C for 3 h in the air atmosphere. After these processes, the samples were prepared.

The purities of samples were tested by X-ray diffraction (XRD) patterns using Rigaku Ultima IV Advanced X-ray diffractometer with a Cu  $K\alpha$  (40.0 KV, 30.0 mA) radiation ( $\lambda$  = 1.5418 Å).







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Photoluminescence excitation (PLE) and emission spectra were measured by Hitachi F-4500 spectrofluorometer equipped with a 150 W Xenon lamp as an excitation source. A cutoff filter was used to avoid the influence of the second-order emission of the source radiation. All the measurements were performed at room temperature.

#### 3. Results and discussion

Fig. 1 shows the typical XRD patterns of CaBi<sub>1.95</sub>B<sub>2</sub>O<sub>7</sub>:0.05Eu<sup>3+</sup> phosphor. One can seen that all of the diffraction peaks can be well fitted with the standard card ICSD #245016 of CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> with a space group P n a 21, consistent with a CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> phase of a orthorhombic structure, featuring the lattice parameters of a = 8.937 Å, b = 5.477 Å, c = 12.591 Å and V = 616.33 Å<sup>3</sup> (Fig. 2 shows the detailed crystal structure of CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub>.). There is no impurity peak detected in the experimental range, suggesting that a pure crystalline compound was obtained. The experimental results indicate that the doped Eu<sup>3+</sup> and charge compensator ions have not led to obvious change in the host structure.

The excitation spectrum ( $\lambda_{em}$  = 591 nm) of CaBi<sub>1.95</sub>B<sub>2</sub>O<sub>7</sub>: 0.05Eu<sup>3+</sup> sample is depicted in Fig. 3(a). It can be found that the excitation spectrum is composed of a broad band at short wavelengths ranging from 200 to 350 nm and a group of sharp bands in the region from 350 to 420 nm. The broad band ranging from 200 to 310 nm is attributed to the overlap between the charge transfer band (CTB) of O<sup>2-</sup>  $\rightarrow$  Eu<sup>3+</sup> [18]. And the group of sharp bands in the region from 350 to 420 nm is consists of a series of sharp peaks at 361 nm, 382 nm, 394 nm and 414 nm, which is belong to the intrinsic 4f–4f transitions of Eu<sup>3+</sup> from the ground state <sup>7</sup>F<sub>0</sub> to the excited states <sup>7</sup>D<sub>4</sub>, <sup>5</sup>G<sub>4</sub>, <sup>5</sup>L<sub>6</sub>, and <sup>5</sup>D<sub>3</sub>, respectively [19]. Fig. 3(b)



Fig. 1. XRD patterns of CaBi<sub>1.95</sub>B<sub>2</sub>O<sub>7</sub>:0.05Eu<sup>3+</sup> phosphor(The standard ICSD#245016 data card of CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> is provided as a reference.).



Fig. 2. Crystal structure of CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub>.



**Fig. 3.** The PL excitation spectrum ( $\lambda_{em}$  = 591 nm) and emission spectrum ( $\lambda_{ex}$  = 394 nm) of a typical CaBi<sub>1.95</sub>B<sub>2</sub>O<sub>7</sub>:0.05Eu<sup>3+</sup> sample.



**Fig. 4.** The emission spectra of CaBi<sub>2-x</sub>B<sub>2</sub>O<sub>7</sub>:xEu<sup>3+</sup> (x = 0.03, 0.05, 0.07, 0.10, 0.13, 0.17 and 0.20) samples excited by 394 nm (Inset: the normalized emission intensity based on the emission intensity of  ${}^{5}D_{0} \rightarrow {}^{7}F_{1}$  transition as a function of Eu<sup>3+</sup> ions concentration).

shows the emission spectra of CaBi<sub>1.95</sub>B<sub>2</sub>O<sub>7</sub>:0.05Eu<sup>3+</sup> sample under the 394 nm excitation. There are four peaks peaking at 581, 591, 614, 655 and 706 nm due to the  ${}^{5}D_{0} \rightarrow {}^{7}F_{j}$  (*j*=0–4) transitions of Eu<sup>3+</sup> ions, respectively [20]. In particular, it can be found that the orange–red emission at 591 nm is dominated.

In order to figure out the optimal doping concentration of  $Eu^{3+}$  ions in the CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> host, a series of CaBi<sub>2-x</sub>B<sub>2</sub>O<sub>7</sub>:xEu<sup>3+</sup> (x = 0.03, 0.05, 0.07, 0.10, 0.13, 0.17 and 0.20) samples were prepared. Fig. 4 shows the emission spectra of CaBi<sub>2-x</sub>B<sub>2</sub>O<sub>7</sub>:xEu<sup>3+</sup> as a function of Eu<sup>3+</sup> concentration excited at 394 nm. It can be found that all the spectra of different contents of Eu<sup>3+</sup> doped CaBi<sub>2</sub>B<sub>2</sub>O<sub>7</sub> are similar and intensity of emission initially increases with the increase of

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