

# Ion dynamics in mechanochemically synthesized amorphous fast ionic conductor $\text{Ag}_2\text{S}-\text{Sb}_2\text{S}_3$

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## Abstract

Frequency dependent electrical conductivity ( $\sigma$ ) of mechanochemically synthesized amorphous fast ion conductors (FICs)  $x\text{Ag}_2\text{S}-\text{Sb}_2\text{S}_3$  ( $x=70, 80, 85$ ) is investigated in the frequency range 5 Hz–13 MHz and temperature range 95–385 K. While the universal dynamic response (UDR) model,  $\sigma(\omega)=\sigma_{\text{dc}}(T)+A(T)\omega^n$ , does reproduce the shape of  $\sigma(\omega)$  spectra, the  $\sigma_{\text{dc}}(T)$  and  $A(T)$  show anomalous characteristics. Both exhibit two distinct Arrhenius regions, a low temperature region with a lower value of activation energy ( $E$ ) and a high temperature region with higher value of activation energy. Further the frequency exponent  $n$  is a (decreasing) function of temperature, decreasing slowly at lower temperatures but more rapidly at higher temperatures. The correlation  $E_{\text{ac}}=(1-n)E_{\text{dc}}$  appears satisfactory in the low temperature region but unsatisfactory at higher temperatures. The isochronal representation of ac conductivity,  $\sigma'$  vs.  $1000/T$  at different frequencies, reveals the existence of a third regime, which is currently under active discussion and is known as superlinear power law (SLPL).

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## 1. Introduction

$\alpha\text{-AgI}$  is the earliest known and yet the best known superionic conductor above  $T_c=148.5^\circ\text{C}$  [1].  $\alpha\text{-Ag}_2\text{S}$  is very similar to  $\alpha\text{-AgI}$  except that it has a higher  $T_c=177^\circ\text{C}$  [2]. Their exceptionally high ionic conductivity ( $\sim 1\ \Omega^{-1}\text{cm}^{-1}$  above  $T_c$ ) is attributed to their unique crystal structure which is variously described as liquid-like or molten-sublattice type in which the mobile cations are in a highly disordered (amorphous) state. During the last three decades, a great deal of effort has been made, with considerable success [3–7] to essentially stabilize this highly disordered, liquid-like and yet crystalline structure [8,9] at room temperature.

Subsequently, various rapid quenching techniques were employed to synthesize noncrystalline or glassy [10–12] fast ion conductors (FICs). Even though  $\alpha\text{-AgI}$  (or  $\text{Ag}_2\text{S}$ ) structure could not be frozen at room temperature all by itself, many FIC glasses have been synthesized in a variety of binary and ternary systems involving  $\text{AgI}$  [13–21] or  $\text{Ag}_2\text{S}$  [22–30] as one of the components. These melt-quenched (MQ) glassy FICs are extensively studied but the issues related to the ion dynamics and various relaxation processes are far from settled [10,31–38].

More recently, high energy ball-milling at room temperature has been successfully used to form amorphous FICs [39–52] which are, by and large, similar to the MQ glasses. However, there are subtle but significant differences between the mechanically-milled (MM) and the MQ amorphous FICs. For instance, the MM glasses generally have a higher conductivity, lower glass transition ( $T_g$ ) and crystallization ( $T_c$ ) temperatures and higher enthalpy ( $\Delta H$ ) and entropy ( $\Delta S$ ) changes at  $T_c$  as compared to MQ glasses. Even more striking is the fact

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that the crystallization in MM glasses generally occurs in two stages [53] while the MQ glasses generally show a single crystallization temperature. Interestingly, the dc conductivity ( $\sigma_{dc}$ ) vs. inverse temperature behaviour of MM glasses is found to exhibit two distinct Arrhenius regions, a low temperature region with a lower activation energy and a high temperature region with a higher activation energy. The existence of two crystallization temperatures in the DSC results and two distinct Arrhenius regions in  $\sigma_{dc}$  vs.  $1/T$  behaviour has given rise to a strong conjecture that there are two distinct bonding states for the  $Ag^+$  ions. Of course, further investigations are desirable to establish these observations.

The frequency dependence of the real part of the complex conductivity

$$\sigma^* = \sigma'(\omega) + j\sigma''(\omega) \quad (1)$$

is extremely useful in understanding the microscopic mechanisms of ion dynamics and various relaxation processes in glassy electrolytes. A large number of FIC glasses have been investigated during the last two decades [54–62]. The ac conductivity ( $\sigma$ ) comprises three different contributions. First, a frequency-independent or dc con-

ductivity ( $\sigma_{dc}$ ), arising from the long-range hopping of the mobile ions, given by

$$\sigma_{dc}(T) = \sigma_0 \exp\left(-\frac{E_{dc}}{kT}\right) \quad (2)$$

where  $\sigma_0$  is the preexponential factor, the thermal activation energy for  $E_{dc}$  conduction and  $kT$  the thermal energy. This dc contribution is invariably present in all glassy FICs and is well understood. Second, a frequency dependent conductivity which obeys a power law is of the type

$$\sigma(\omega, T) = A(T)\omega^n \quad (3)$$

where  $A(T)$  is independent of frequency and has Arrhenius temperature dependence

$$A(T) = A_0 \exp\left(-\frac{E_{ac}}{kT}\right) \quad (4)$$

where  $A_0$  is a constant for a material and  $E_{ac}$  is called ac activation energy. The frequency exponent  $n$  is supposedly a constant ( $0 < n < 1$ ).

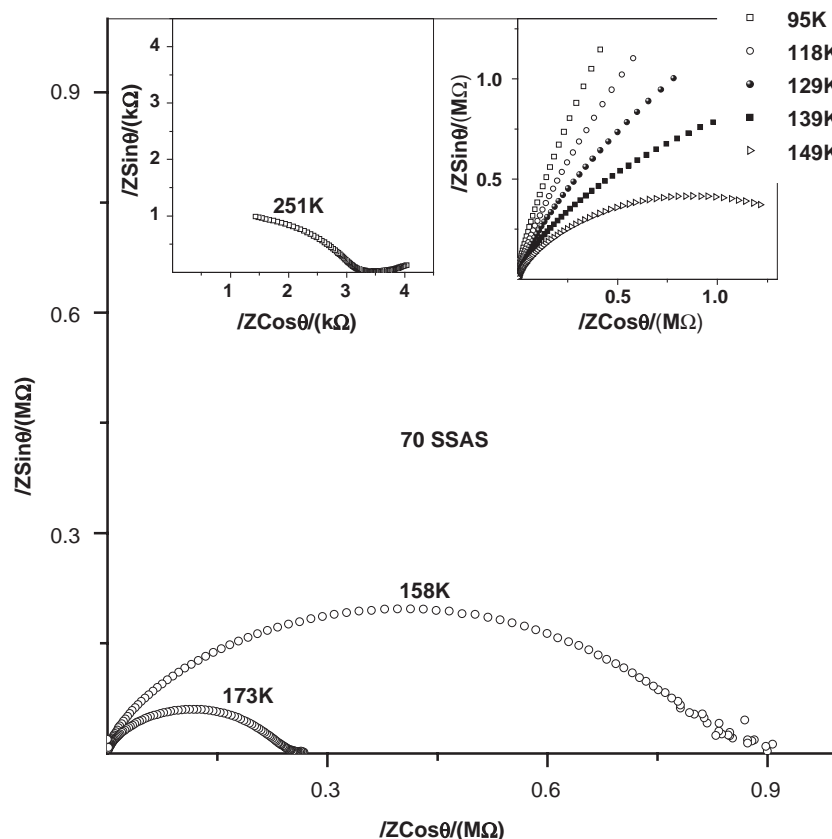


Fig. 1. Impedance plots for the mechanochemically synthesized  $70Ag_2S-Sb_2S_3$  (70 SSAS) amorphous fast ion conductor at different temperatures. Insets of the plots show the temperatures for which complete spectra are not accessible.

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