

Effects of seeding methods on the fabrication of microcrystalline silicon solar cells using radio frequency plasma enhanced chemical vapor deposition

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Abstract

Single junction *p-i-n* $\mu\text{c-Si:H}$ solar cells were prepared in a low-cost, large-area single chamber radio frequency plasma enhanced chemical vapor deposition system. The effects of seeding processes on the growth of $\mu\text{c-Si:H}$ *i*-layers and performance of $\mu\text{c-Si:H}$ solar cells were investigated. Seeding processes, usually featured by highly hydrogen rich plasma, are effective in inducing the growth of $\mu\text{c-Si:H}$ *i*-layers. It has been demonstrated that *p*-layer seeding methods are preferable to *i*-layer seeding. While performance of $\mu\text{c-Si:H}$ solar cells produced by *i*-layer seeding methods was usually limited by very low fill factors, $\mu\text{c-Si:H}$ solar cells with good initial and stabilized conversion efficiencies were obtained by *p*-layer seeding.

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1. Introduction

Over the past decade, advances have been made in the fabrication of solar cells with hydrogenated microcrystalline silicon ($\mu\text{c-Si:H}$) intrinsic (*i*-) layers. High quality $\mu\text{c-Si:H}$ is usually deposited by plasma enhanced chemical vapor deposition (PECVD). Early success in fabricating $\mu\text{c-Si:H}$ solar cells was achieved by *very high frequency PECVD* [1]. Recently, studies on the fabrication of $\mu\text{c-Si:H}$ solar cells using conventional radio frequency (RF-) PECVD have attracted extensive attention due to its compatibility with the existing large-scale manufacturing technology of amorphous silicon ($\alpha\text{-Si:H}$) photovoltaic (PV) modules. In most laboratory scale research, sophisticated, multi-chamber, load-locked deposition systems, expensive laboratory substrates, as well as highly effective back reflector or transparent interlayer were routinely used to obtain high

efficiency $\mu\text{c-Si:H}$ solar cells [2–4]. However, the true potential of $\mu\text{c-Si:H}$ solar cells can only be evaluated under conditions of cost-competitive manufacturing of large-area PV modules. Industrial production of 8 Ft² $\alpha\text{-Si:H}$ PV modules by a massively-parallel batch process in single chamber RF-PECVD systems has been successfully developed [5]. A realistic approach based on this proven technology has been taken to study the manufacturing of $\mu\text{c-Si:H}$ solar cells on a low-cost, large-area basis [6]. Generally, glow discharge of highly hydrogen diluted silane is used as the standard PECVD approach for $\mu\text{c-Si:H}$ deposition. Although the microscopic mechanism of $\mu\text{c-Si:H}$ growth has yet to be better understood, it is generally believed that the initial nucleation of $\mu\text{c-Si:H}$ crystallites on $\alpha\text{-Si:H}$, $\alpha\text{-SiC:H}$ or other under-layers is critical to obtaining high quality $\mu\text{c-Si:H}$ films. In this study, therefore, a wide variety of seeding processes were systematically explored and the effects of seeding schemes on the growth of $\mu\text{c-Si:H}$ and device performance of $\mu\text{c-Si:H}$ solar cells were investigated.

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2. Experimental details

Single junction *p-i-n* solar cells with $\mu\text{c-Si:H}$ *i*-layers (absorbers) were fabricated on commercial grade SnO_2 superstrates by RF-PECVD method (13.56 MHz), using silane highly diluted by hydrogen as feed gas mixtures. Substrate temperature was usually kept near 200 °C. This single chamber deposition system, without load-lock, was routinely exposed to air for unloading and loading the substrates. The advantages of this low-cost PECVD system include a large electrode utilization ratio (a single 12 in. $\times$ 15 in. powered electrode was used to simultaneously coat four substrates equal to its size), high gas utilization, a controllable contamination profile, ease of operation, and low maintenance. The doped *p*-layers and *n*-layers, as well as *i*-layers, were deposited in the same reactor without any movement of the substrates and/or the reactor. A thin $\alpha\text{-SiC:H}$ *p*-layer was first deposited on the SnO_2 superstrate. Then, a seeding step was applied to induce crystallization for the Si:H *i*-layer followed by the deposition of bulk *i*-layer. Finally, an $\alpha\text{-Si:H}$ *n*-layer was grown. Sputtered Al, without any rear reflection enhancement schemes, was used as the standard back contact.

Device fabrication and performance testing, including current–voltage curve (*I*–*V*), spectral response (QE), and light soaking, were routinely conducted at Energy Photovoltaics, Inc. (EPV). Parameters obtained from performance testing, such as conversion efficiency, open circuit voltage (*V*_{oc}), short circuit current density (*J*_{sc}), fill factor (FF), and red-light spectral response (QE at 800 nm or longer), were also used to deduce the microstructural properties of $\mu\text{c-Si:H}$ solar cells. In particular, low *V*_{oc} and high red-light response were used as the signatures of $\mu\text{c-Si:H}$ *i*-layers. Raman Spectroscopy of actual solar cells, performed at New Jersey Institute of Technology (NJIT), was used to study the micro-crystallinity of $\mu\text{c-Si:H}$ solar cells. Due to the contribution from substrates and vary optical properties of Si:H *i*-layers, the micro-crystallinity was presented in terms of the ratio of peak intensities (*I*_c/*I*_a) of Raman shift corresponding to $\mu\text{c-Si:H}$ (*I*_c) and $\alpha\text{-Si:H}$ (*I*_a), respectively.

3. Results and discussion

3.1. Seeding methods for $\mu\text{c-Si:H}$ *i*-layer deposition

The performance of $\mu\text{c-Si:H}$ solar cells depends on many processing details, chief among which are the seeding steps and the growth conditions for the bulk *i*-layers. Throughout this study, it has been established that, for a fixed set of *i*-layer plasma conditions capable of sustaining $\mu\text{c-Si:H}$ growth, the seeding or incubation procedure (which may comprise several individual steps) largely determines the properties of Si:H absorber, and

strongly influences the device performance. Under the same *i*-layer growth conditions, amorphous, mixed-phase, or micro-crystalline (nano-crystalline) Si:H absorbers can be obtained, respectively, depending on the seeding method.

The seeding methods we have explored can be classified into two categories: (i) *p*-layer seeding, which refers to all seeding methods involving boron doped *p*-layer, and (ii) *i*-layer seeding referring to the nucleation methods inside the ‘intrinsic’ Si:H layer. To grow $\mu\text{c-Si:H}$ on an $\alpha\text{-Si:H}$ or $\alpha\text{-SiC:H}$ under-layer, a defective transition layer may exist at or near the *p/i* interface which may severely affect device performance. Thus, its thickness should be limited. Highly hydrogen rich plasma conditions are usually applied during seeding steps to induce initial $\mu\text{c-Si:H}$ nucleation followed by growth of bulk $\mu\text{c-Si:H}$ *i*-layer under relatively softer plasma conditions. However, the hydrogen rich plasma used in seeding processes could do severe damages to the microstructure and performance of $\mu\text{c-Si:H}$ solar cells. Thus, the *p*-layer seeding approaches take the advantage of limiting the damages associated with energetic seeding plasma within the PV non-active *p*-layer and consequently lead to better overall carrier collection. Conceptually, *i*-layer seeding methods take the advantages of minimized optical loss associated with the thicker, defective *p*-layers resulting from *p*-layer seeding approaches and higher *V*_{oc} due to the wider band gap of the non-microcrystalline *i*-layer near the *p/i* interface. Presumably, the disadvantage of this type seeding methods is the unavoidable defects near the *p/i* interface (*i*-layer side) created by the hydrogen rich plasma used to create nucleation sites, which might lead to poor carrier collection.

The *i*-layer seeding processes usually consist of the deposition of a thin $\alpha\text{-Si:H}$ buffer layer, an incubation layer deposited by pure hydrogen etching on the buffer layer or seeding using very high hydrogen dilution ratio, and a silane grading step leading to the growth of bulk $\mu\text{c-Si:H}$ *i*-layer. The *p*-layer seeding processes consist of similar approaches with $\mu\text{c-Si:H}$ nucleation occurs within doped *p*-layer.

3.2. Effects of seeding methods on the growth of $\mu\text{c-Si:H}$

High hydrogen to silane dilution ratio has been widely recognized as the most important factor to induce and sustain the growth of $\mu\text{c-Si:H}$. Therefore, hydrogen to silane dilution ratio, $R=[\text{H}_2]/[\text{SiH}_4]$, was used as the major parameter in analyzing the seeding processes. In the relatively large area RF-PECVD system, it has been demonstrated that the seeding schemes so far explored, such as pure hydrogen etching on $\alpha\text{-SiC:H}$ or $\alpha\text{-Si:H}$ buffer layer and seeding by very high hydrogen dilution ratio, are effective in inducing the formation of $\mu\text{c-Si:H}$. Without special seeding methods, high hydrogen dilution ratio is necessary for the growth of bulk $\mu\text{c-Si:H}$ *i*-layers.

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